DEVELOPMENT OF AVENTURINE GLAZES

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Aventurine glazes, using Pakistani raw materials in their formulation have been developed. The covered(a) variation in RO group (Na,K,Ca and Pb), (b) variation in R₂O₃ group (B₂O₃, Al₂O₃ and Fe₂O₃), (c) variation in RO₂ (SiO₂) and effects of application and firing techniques of glazes. The glazes were fired up to 1000° C. Some good Aventurine glazes (containing Fe) were obtained and conditions that affect the results were also established.

Aventurine quartz and aventurine felspar have scales of mica, haematite and gothite, which give off fire-like flashes when held in bright lights. For the first time in 1849, Wohler¹ produced such effects in glazes which resembled aventurine felspar, and these glazes were therefore named aventurine glazes.

Aventurine glaze is one in which individual crystal in the magma separates out and appears to be flitter or spangle suspended in the glass. The aventurine effect can be obtained by adding a metallic oxide in such quantities that when heated to a molten mass, the matrix becomes saturated with the oxide; and when the glaze cools thin crystals or spangles of the oxide separate out from the supersaturated magma. The commonly used oxides for this purpose are those of iron, chromium and copper. Iron oxide is more common and gives the best aventurines.

Many workers have made investigations in this field. An aventurine effect by applying an engobe of iron or uranium oxide and then covering the same with a borax glaze.¹ Thirty-five parts of iron oxide per 100 parts of the glaze, and 20% finely-ground metallic iron produced aventurine glazes with variations in colour.² The addition of artificial haematite crystals to the glaze produces the same effect.³ High amounts of SiO₂ require high amounts of Fe₂O₃, as high as 0.71 moles.⁴ These glazes were studied rather extensively by making changes in RO and R₂O₃ groups while the SiO₂ contents were kept constant.⁵ In another study,⁶ iron was substituted by Cr and Cu. All these glazes were fired between cone 02 to cone 3.

In the present investigation, the authors have not only changed the limits of ranges of RO and R_2O_3 which the previous research workers had covered but have also included the study of effects of variations in $RO_2(SiO_2)$.

Experimental

Plan of Investigation.—The investigation consists of four systematic fields i.e. effects of (1) variation

in B_2O_3 and Fe_2O_3 ; (2) variation in Al_2O_3 content; (3) variation in RO group (Na,K, Ca, Pb);(4) variation in RO₂ group (SiO₂). Effects of firing conditions and glaze application have also been studied.

From previous investigations it was found that. the formula:

$$\begin{array}{cccc} 1 & \text{Na}_2\text{O} & 0.15 & \text{Al}_2\text{O}_3 \\ & 0.75 & \text{Fe}_2\text{O}_3 & 7.0 & \text{SiO}_2 \\ & 1.25 & \text{B}_2\text{O}_3 & \end{array}$$

gave the best composition for the aventurines. In preliminary investigation Fe_2O_3 was varied from 0.75 to 0.95 mole, keeping other constituents constant. It was found that 0.95 mole Fe_2O_3 gave dark coloured glaze without crystallisation, while 0.75 mole, gave yellow coloured glaze with crazing. The rest of the glazes gave crystals, although the colour developed darker shades with increase in Fe_2O_3 contents. Keeping in view all the previous investigations the following ranges of various constituents were selected as the base:

0.4- 1 Na₂O 0.05-0.20 Al₂O₃ 3.5-11 SiO₂ 0.0- 0.6 K₂O 0.75-0.90 Fe₂O₃ 0.0- 0.6 CaO 1.25-2.25 B₂O₃ 0.0- 0.6 PbO.

In the first series, Fe_2O_3 and B_2O_3 were varied as shown in Fig. 1. Ascending alphabets represent rise in B_2O_3 while rising numbers show the respective rise in Fe_2O_3 contents. Batch compositions so formulated are given in Tables 1a and 1b. The two best compositions C_4 and D_1 were selected from these series and variation in Al_2O_3 content was made. Batch compositions are given in Table 2.

Figure 2, the triaxial, shows the variation in RO group. Since Pb and Ca have very little favourable effect upon crystallisation, while Na_2O , is most suitable, 0.4 mole of Na_2O was kept constant and the remaining 0.6 mole of RO group was replaced by K_2O , CaO, and PbO. Ten compositions as shown in Tables 3a and 3b were

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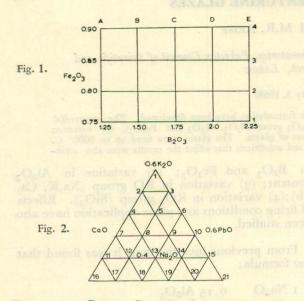


TABLE 12.—BATCH COMPOSITIONS FOR THE FRITS IN I ST SERIES.

	Frit	0/	0/ 5	% odium	0/	0/	0/	were 1000
	No.	Felspar	Borax	car-	Quartz	% Fe ₂ O ₃	% Boric	of th
	- Change	assister of	ano grat	onate	and the	+1 1	acid	
	A-1	8.5	40.5	5.4	25.0	20.4	0.0	Te
	B-1	8.1	38.6	5.1	23.8	19.4	5.0	The
	C-1 D-1	7.7 7.35	36.6 35.00	4.9 4.65	22.6 21.6	18.5	9.5 13.7	in m
	E-1	7.0	33.6	4.03	20.7	17.6 16.9	17.4	follow
	A-2	8.4	40.0	5.3	24.7	21.4	0.0	
	B-2	8.0	38.0	5.1	23.4	20.4	5.0	1-
	-C-2	7.7	36.2	4.8	22.4	19.4	9.4	(1)
	D-2	7.4	34.7	4.6	21.3	18.5	13.5	
	E-2	7.0	33.2	4.4	20.4	17.8	17.3	
	A-3	8.3	39.4	5.2	24.4	22.5	0.0	
	B-3	7.9	37.6	5.0	23.2	21.4	4.9	(2
	C-3 D-3	7.5 7.2	35.8 34.2	4.8	22.1 21.1	20.4	9.3	0.0311
	E-3	6.9	32.8	4.3	20.2	19.5 18.7	13.4 17.0	
	A-4	8.2	39.0	5.5	24.0	23.6	0.0	
	B-4	7.8	37.0	5.0	23.0	22.4	4.8	
	C-4	7.5	35.5	4.6	21.8	21.4	9.2	
	D-4	7.1	33.8	4.5	20.8	20.4	13.4	(3
	E-4	6.8	37.5	4.3	20.0	19.6	16.9	()
	CHOME.	Subsection of the	an service	TAR	F th	MILTI	BATCHES	OF TH
_				I ADI		WIILL J	DATCHES	OF II
	No.	Fri	it %	Clay	7 %	Quar	tz %	No.
	A-1	6=	5.3	. г.	67	33.	0.0.1/	A-3
	В-1		5.0		62	32.		B-3
	C-I		5.8		58	31.	6	C-3
	D-I		7.8					D
					53	30.		D-3
	E-I	50	3.2	Ι.	5	30.	2	E-3

formulated from three frits K1,K7 and K10. Base glaze composition was as shown at C_4 .

In case of variation in the SiO₂ content, C₄ again was taken as base composition and additions of SiO₂ were made in mill batch. The compositions formulated are shown in Table 4.

Preparation of Frit and Glaze.-Frit batches, according to the compositions shown in the tables, were weighed and thoroughly mixed, smelted in grog crucibles and quenched in water. The frit thus formed was dried and ground in an edgerunner up to 100 mesh. Glaze batches were formulated by adding frit clay, quartz and strong ammonium hydroxide as deflocculant. The glaze was ground in pebble mill up to a fineness of 200 mesh.

Glaze Application and Firing.-Small test pieces were cast in the form of crucibles, ash trays and figurines etc. from a slip having a composition of clay 55%, felspar 30%, and quartz 15%. Glazes were applied on the pieces already biscuited at o°C by dipping as well as spraying. Thickness ne glaze varied from 1 mm to 2 mm.

est pieces were fired in electric muffle furnace. maximum temperature attained was 1000°C nost of the cases. Cooling was done in the wing way:-

- () Test pieces were soaked for 30 to 45 min and then the furnace was allowed to cool in 16 hr.
- 2) Test pieces were soaked for $\frac{1}{2}$ hr controlled cooling was done to 800°C in 4 hr and then the furnace was allowed to cool in 14 hr.
- The above was repeated with controlled 3) cooling down to 750°C.

No.	Frit %	Clay %	Quartz %	No.	Frit %	Clay %	Quartz %
А-1	65.3	1.67	33.0	A-3	66.0	1.6	32.4
B-1	66.0	1.62	32.3	B-3	66.8	1.6	31.6
C-I	66.8	1.58	31.6	C-3	67.7	1.55	30.8
D-I	67.8	1.53	30.45	D-3	68.3	1.5	30.2
E-1	68.2	1.5	30.2	E-3	69.0	1.48	29.5
A-2	66.0	1.6	32.2	A-4	66.5	1.6	32.I
B-2	67.0	1.6	31.2	B-4	67.0	1.6	31.3
C-2	67.8	1.5	30.7	C-4	67.9	1.53	30.6
D-2	68.6	1.5	29.8	D-4	68.5	1.5	30.0
Е-2	68.7	1.5	29.8	E-4	69.3	1.46	29.20

HE GLAZES IN I SERIES.

Swine ISPA	U. Ser ita Dina te	JEKIES.	au ussa	81 (2121223)
No.	Mol. eq. of Al_2O_3	Frit %	Clay %	Quartz %
IC4	0.I	69	a zoa ea In e rri d	31
2C4	0.15	67.9	1.53	30.6
$3C_4$	0.20	67.0	3.0	30.2
$4C_4$	0.25	66.0	4.5	24.6
IDI	0.I	69.0	own.	31
$2D_{I}$	0.15	67.8	1.53	30.4
$3D_{I}$	0.20	66.6	3.0	30.4
4D ₁	0.25	65.5	$4 \cdot 5$	30.0

TABLE	2.—BATCH	Composition	OF	GLAZES	IN
		II SERIES.			

TABLE 3a.—BATCH COMPOSITIONS OF FRITS IN III SERIES.

No.	Felspar	Borax	Boric acid	SiO2	Fe ₂ O ₃	K ₂ CO ₃	CaCO3	Pb ₃ O ₄
K-1	4.8	19.5	19.10	21.10	23.3	12.2		
K-7	4.9	20.2	19.7	24.2	21.9	10 <u>.1</u> 094	9.1	79 <u>(1</u>
K-10) 4.4	18.2	17.6	21.6	19.6	Wr ed R.	. guile	18.6

TABLE 3D.—BATCH COMPOSITION OF GLAZES IN III SERIES.

No.	Frit K-1	Frit K-7	Frit K-10	Clay	Quartz
K-1	66.0	an el se n el	1 10	3.5	30.2
K-2	44.6	21.2		3.48	30.7
K-3	42.9	199 (0 <u>-0</u> 7	24.4	3.3	29.4
K-4	22.6	43.0	itros <u>pa</u> rri	3.5	31.0
K-5	21.6	20.5	24.5	3.4	31.0
K-6	20.8		47.3	3.3	28.7
K-7		65.0		3.5	31.5
K-8	es <u>ur</u> tes hai	41.6	24.8	3.4	30.0
K-9		20.0	47.7	3.3	29.0
K-10		-	69.0	3.2	27.8

TABLE 4.—BATCH COMPOSITIONS OF THE GLAZES IN IV SERIES.

No.	Moles of SiO ₂	Frit C ₄ %	Quartz %	Clay %
S-1	3.5	93.5	4.3	2.2
S-2	4.0	88.6	9.3	2.1
S-3	4.5	84.5	13.6	2.2
S-4	5.0	80.5	17.5	1.9
S-5	5.5	77.0	21.4	г.8
S6	6.0	73.5	24.8	1.7
S-7	6.5	70.5	27.8	1.6
S8	7.0	68.0	30.6	1.6
S-9	7.5	63.3	33.3	1.5
S-10	8.0	63.0	35.6	I.4
S-II	9.0	58.6	40.0	I.4
S-12	10.0	54.9	43.8	1.3
S-13	11.0	51.5	47.2	I.2

- (4) Test pieces were soaked at 1000°C for ¹/₂ hr and then controlled cooling was done up to 800°C in 1 hr and again at this temperature, the test pieces were allowed to cool down to room temperature in 13 hr.
- (5) Test pieces were fired at 1000°C and were allowed to cool without any soaking.
- (6) Test pieces were soaked at 950°C (maximum) and controlled cooling was done to 750°C in 3 hr and then furnace was cooled to room temperature in 13 hr.

Discussion

From the observations in Table 5, it is found that the lower amounts of B_2O_3 and $Fe_2O_3(A-1,$ A-2 and B-1) give yellow colour to the glaze, with lot of crazing. When either of the two is increased, a marked increase in crystal growth is observed. Higher amounts of Fe_2O_3 increase the maturing temperature of the glazes and the development of crystals. This is evident from the compositions A-3 and A-4 which have considerable Fe_2O_3 melt at higher temperature as compared to A-1 and A-2. Because of higher percentage of Fe_2O_3 , the colour of the glazes darkens, crystals become more prominent and uniform, and crazing is minimised.

When B_2O_3 is increased, crystal growth is enhanced again, colour of the glaze darkens but there is no appreciable change in the melting temperature of the glazes. D_I and E_I prove these conclusions.

In between these limits, the results are as expected, i.e. B-2, C-3 and C-4 show gradual difference in colour and crystal formation. In fact these compositions are amongst the best ones. Still higher amounts of B_2O_3 and Fe_2O_3 do not give good results. Best aventurines are obtained in the ranges, B_2O_3 from 1.50 to 2.0 moles and Fe_2O_3 from 0.80 to 0.90 mole.

Alumina, when varied from 0.10 to 0.25 mole gives best results when it is 0.10 mole. Higher alumina does not favour crystallisation.

When Na₂O is replaced by K_2O , CaO and PbO(Fig.2) crystal growth is minimised but the glaze exhibits many pleasant shades of colours. K_1 gives a chocolate colour. This colour becomes lighter and lighter as we move towards PbO apex (Fig. 2) i.e. K_6, K_9 and K_{10} give the lightest shades. A pleasant bluish tinge is imparted when we move towards CaO, i.e. K_4, K_7 and K_8 are

TABLE 5.—RESULTS.

Glaze No.	Colour	Crazing	Crystalli- zation	Approxi- mate melting temp.°C
A-1 B-1 C-1 D-1	Yellow-purple Purple	crazed not visible	None fair "	980 ,, ,,
E-1 A-2 B-2 C-2	" Yellow-purple Purple	" crazed not visible "		,, 990 ,, ,,
D-2 E-2 A-3 B-3 C-3	Dark red	55 35 35 35 35 35 35 35 35	good "	" 1000 "
D-3 E-3 A-4 B-4	" , Red-black ,	" " " "	very good good very good	>> >> >> >> >> >> >> >> >> >>
C-4 D-4 E-4 1C-4 2C-4	" " Red-black	" " crazed not visible	excellent good fair very good	33 33 33
3C-4 4C-4 K-1 K-2	" Purple Purple with bluish tint	" not visible	good none fair	
К-3 К-4	Purple with Brown tint Purple with more bluish	not visible not visible	very little very little	inimised. When B
К-5 К-6 К-7	tint Purple Purple with brownish tint Dominantly bl		esan, colou e apprecial e of, the gia	
K-8 K-9 K-10	purple Lighter bluish tint Purple	" not visible	none "	
$\left.\begin{array}{c} S-1\\ S-2\\ S-3\end{array}\right\}$	Brownish tint Purple		", fair	icse comp till higher ive-good
S-4 S-5 S-6	0 2 0 100155 6. 8. 80 60 9 75 1		very good	
S-7 S-8 S-9	K ₂ O CrO	none	very good	
S-10 S-11 S-12 S-13	national de la colorada de la colora		good	

darker; K_7 being most bluish in colour. No crazing is seen in these glazes and all of them have good gloss and smooth surface.

Variation in SiO_2 content from 3.5 moles to 11.0 moles has revealed that low SiO_2 contents give very little crystals. The number of crystals increases as we increase SiO_2 . 7.0 to 9.0 moles of SiO_2 give the optimum range for best crystals, while SiO_2 as high as 11.0 moles does not hinder crystal growth.

Application of these glazes requires special attention. Skilled workmanship is essential. Uneven coating or any streak on ware spoils the uniformity of the crystals. Thin application of the glazes does not give rise to crystals because the glaze is not sufficient in which crystals can develop. Thick application gives rise to flowing and uneven crystal growth. I mm to 2 mm thickness gives good results.

Development of crystals depends upon the rate of cooling. Slow cooling between 1000°C and 750°C is most favourable for crystal growth, when cooling was done as in firing at Nos. 2,3 and 4 mentioned under para 'Glaze application and firing'; the results were not different from one another. In case of firing No. 1, some crystals were observed. In case of firing No. 5, no crystallisation was noted, as the glaze got less time to melt and form crystals. No. 6, again gave good results. These firing confirm the results of Norton4, on control of crystalline glazes.

The glaze which gives the best results has the following composition.

1.0 Na ₂ O	0.15 Al ₂ O ₃	7.0 SiO2
	0.90 Fe ₂ O ₃	
	I.75 B ₂ O ₃	

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